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# Magnetic and electrical transport properties of $Gd_2CoGe_6$ and $Tb_2CoGe_6$ germanides

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## ABSTRACT

The low-temperature physical properties of two intermetallics, Gd<sub>2</sub>CoGe<sub>6</sub> and Tb<sub>2</sub>CoGe<sub>6</sub>, crystallizing with the orthorhombic Ce<sub>2</sub>CuGe<sub>6</sub>-type structure, have been studied by means of magnetization and electrical resistivity measurements. The magnetic behavior of these compounds involves the presence of magnetic moments on the rare earth and cobalt atoms sites. Both phases have been found to order antiferromagnetically below 20 and 22 K, respectively. In each case, complex behavior of the magnetic susceptibility in the ordered state hints at some changes in the magnetic structure on decreasing the temperature. The two compounds exhibit metallic-type electrical conductivity.

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## 1. Introduction

Homologous series of rare-earth (R) intermetallics with the general composition  $R_2TGe_6$  have been reported to form with various d-electron transition metals T = Cu, Ag, Au, Pd, Pt [1], Ni [2], and Co [3]. These phases crystallize with an orthorhombic crystal structure of the Ce<sub>2</sub>CuGe<sub>6</sub>-type (space group *Amm2*), which can be described as an ordered stacking of the AlB<sub>2</sub>-, BaAl<sub>4</sub>-,  $\alpha$ Pu-, and ZrSi<sub>2</sub>-type slabs [4]. Isotypic ternary germanide Ce<sub>2</sub>MnGe<sub>6</sub> was found in the Ce–Mn–Ge system [5], and recently the crystal structure and magnetic properties investigations were reported for La<sub>2</sub>MnGe<sub>6</sub> [6]. Most of the hitherto studied R<sub>2</sub>TGe<sub>6</sub> compounds order antiferromagnetically at low temperatures [2,7–13]. Sometimes they show fairly complex magnetic behavior, possibly related to the presence of two crystallographically inequivalent sublattices of magnetic R atoms.

In this paper we report for the first time on the physical behavior of two isotypic compounds, namely Gd<sub>2</sub>CoGe<sub>6</sub> and Tb<sub>2</sub>CoGe<sub>6</sub>.

#### 2. Experimental details

Polycrystalline samples of Gd<sub>2</sub>CoGe<sub>6</sub> and Tb<sub>2</sub>CoGe<sub>6</sub> were synthesized by arcmelting the elemental constituents (nominal purities: Gd, Tb – 99.9 wt.%, Co – 99.99 wt.%, Ge – 99.999 wt.%) on a water-cooled copper hearth in titanium-gettered argon atmosphere. The ingots were turned over and re-melted a few times to ensure good homogeneity. Subsequently, the buttons were sealed in evacuated quartz tubes and annealed at 600 °C for two months. Finally the products were quenched in cold water.

\* Corresponding author. E-mail address: D.Kaczorowski@int.pan.wroc.pl (D. Kaczorowski). The sample quality was examined by X-ray powder diffraction at room temperature on a DRON-2.0 m powder diffractometer using Fe  $K_{\alpha}$  radiation. The phase analysis was carried out and the lattice parameters were determined using the CSD software package [14].

Magnetic measurements were performed in the temperature range 1.72–400 K and in external fields up to 5 T using a Quantum Design MPMS-5 SQUID magnetometer. The electrical resistivity was measured over the temperature interval 4.2–300 K employing the standard dc four-probe technique.

#### 3. Results

The X-ray powder diffraction measurements revealed that the two compounds investigated crystallize with orthorhombic unit cells of the space group *Amm2*. The lattice parameters derived from the X-ray pattern of Gd<sub>2</sub>CoGe<sub>6</sub> are a = 0.3956(4) nm, b = 0.4016(4) nm and c = 2.1495(2) nm, while those refined for Tb<sub>2</sub>CoGe<sub>6</sub> are a = 0.3938(1) nm, b = 0.4006(1) nm and c = 2.1408(4) nm. No foreign Bragg peaks were observed, hence indicating single-phase character of the obtained samples.

The magnetic properties of Gd<sub>2</sub>CoGe<sub>6</sub> and Tb<sub>2</sub>CoGe<sub>6</sub> are displayed in Fig. 1. Above 20 K, the inverse magnetic susceptibility of the former compound exhibits a straight-line behavior that can be well described by the Curie–Weiss (CW) formula,  $\chi_m(T) = C/(T - \theta_p)$ , where *C* is the Curie constant, and  $\theta_p$  is the paramagnetic Curie temperature. The least-squares fitting to the experimental data yielded *C* = 17.47 (emu K)/mol and  $\theta_p$  = -8.0 K. Similar approximation of the  $\chi_m(T)$  curve of Tb<sub>2</sub>CoGe<sub>6</sub> is possible only above about 100 K, and the so-derived parameters are *C* = 25.03 (emu K)/mol and  $\theta_p$  = -13.8 K. The limited applicability of the CW law to the latter compound likely arises from crystalline electric field (CEF) interactions, which cause splitting of the 13-fold degenerated <sup>7</sup>*F*<sub>6</sub> ground



**Fig. 1.** Temperature dependence of the reciprocal magnetic susceptibility of (a)  $Gd_2CoGe_6$  and (b)  $Tb_2CoGe_6$ . The solid lines represent the Curie–Weiss fits described in the text. The upper insets show the magnetic susceptibility at low temperatures. The lower insets present the field dependence of the magnetization taken at 1.72 K with increasing (full circles) and decreasing (open circles) magnetic field strength.

multiplet of Tb<sup>3+</sup> ions in the electrostatic potential due to neighboring atoms. Such an effect is not observed for Gd<sub>2</sub>CoGe<sub>6</sub> because trivalent gadolinium ions have the angular momentum L=0 (the ground multiplet  ${}^{8}S_{7/2}$  can be split in magnetic field only).

For both compounds, the derived Curie constant *C* is distinctly larger than the respective value calculated with the Russell–Saunders *L–S* coupling scenario for free Gd<sup>3+</sup> and Tb<sup>3+</sup> ions. This finding signals that the magnetic properties of Gd<sub>2</sub>CoGe<sub>6</sub> and Tb<sub>2</sub>CoGe<sub>6</sub> are probably governed not only by the magnetic moments carried on the rare-earth ions but also by the magnetic moments of cobalt. Setting  $C = (1/8)[2(\mu_{eff}^R)^2 + (\mu_{eff}^{Co})^2]$  and assuming that the rare-earth contributions (the first term) are equal to the theoretical *L–S* values, i.e.  $\mu_{eff}^{Gd} = 7.94 \,\mu_B$  and  $\mu_{eff}^{Tb} = 9.72 \,\mu_B$ , one obtains the effective magnetic moment due to cobalt (the second term) to be 3.70  $\mu_B$  in Gd<sub>2</sub>CoGe<sub>6</sub> and 3.36  $\mu_B$  in Tb<sub>2</sub>CoGe<sub>6</sub>. These values can be compared with the spin-only effective moment of Co<sup>2+</sup> ion ( $\mu_{eff}^{Co} = 3.87 \,\mu_B$ ) with the electronic ground state  ${}^{4}F_{9/2}$ . Though the experimental values of  $\mu_{eff}^{Co}$  are somewhat smaller than the theoretical one, the contributions due to cobalt to the magnetic susceptibility of the two ternaries are instantly recognizable.

The negative values of the paramagnetic Curie temperatures in  $Gd_2CoGe_6$  and  $Tb_2CoGe_6$  hint at significant antiferromagnetic correlations that eventually may give rise to long-range magnetic ordering. Indeed, as demonstrated in the upper insets to Fig. 1a and b, respectively, for both compounds, the magnetic susceptibility  $\chi_m(T)$  shows a maximum that signals the onset of antiferromagnetic state. The so-defined Neel temperatures  $T_N$  are 20 K and 22 K for Gd<sub>2</sub>CoGe<sub>6</sub> and Tb<sub>2</sub>CoGe<sub>6</sub>, respectively. In the ordered state, the magnetic susceptibility does not decrease with decreasing temperature, as would be expected for a simple antiferromagnet. Instead, below a shallow minimum near 15 K, the  $\chi_m(T)$  variation of Gd<sub>2</sub>CoGe<sub>6</sub> exhibits a monotonic upturn, while that of Tb<sub>2</sub>CoGe<sub>6</sub> shows a kink at 10 K (the little feature seen near 2 K is probably due to tiny admixture of Tb<sub>2</sub>O<sub>3</sub> [15]). This fairly complex behavior seen for each compound may suggest some changes in their magnetic structures occurring on decreasing the temperature.

The antiferromagnetic nature of the ordered state in Gd<sub>2</sub>CoGe<sub>6</sub> and Tb<sub>2</sub>CoGe<sub>6</sub> is reflected in characteristic shapes of the magnetization isotherms taken at 1.72 K (see the lower insets to Fig. 1). In weak magnetic fields, the  $\sigma(H)$  curve of the Gd-based material exhibits a straight-line behavior up to about 1.5 T, at which field a spin-flop transition takes place, i.e. the antiparallel magnetic moments reorient perpendicular to the applied field. Similar effect, yet much better pronounced, is observed for Tb<sub>2</sub>CoGe<sub>6</sub> near  $\mu_0 H$  = 3.2 T. Another feature revealed for the latter phase is some small spontaneous magnetization seen in weak fields and little hysteresis in  $\sigma(H)$  noticeable in strong fields. Both effects hint at the presence of tiny ferromagnetic contribution to the magnetization at 1.72 K. Attributing this weak ferromagnetic signal to an intrinsic behavior of Tb<sub>2</sub>CoGe<sub>6</sub> it might be considered as an evidence for canted or non-collinear antiferromagnetic arrangement of the terbium magnetic moments at this temperature. Another possible explanation involves some small ferri- or ferromagnetic component due to the cobalt sublattice.

Fig. 2 shows the temperature dependencies of the electrical resistivity of  $Gd_2CoGe_6$  and  $Tb_2CoGe_6$ . Both compounds exhibit metallic character of the electronic transport, reflected in the magnitudes of the resistivity measured at room temperature and the overall shapes of the  $\rho(T)$  curves. In the paramagnetic region, the resistivity data of  $Gd_2CoGe_6$  can be well approximated by the Bloch–Grüneissen–Mott (BGM) formula [16]

$$\rho(T) = (\rho_0 + \rho_0^{\infty}) + 4RT \left(\frac{T}{\Theta_R}\right)^4 \int_0^{\Theta_R/T} \frac{x^5 dx}{(e^x - 1)(1 - e^{-x})} - KT^3$$

where the first term represents the scattering of conduction electrons on static defects in the crystal lattice (residual resistivity  $\rho_0$ ) and on disordered magnetic moments (spin-disorder resistivity  $\rho_0^{\infty}$ ), the second term describes the phonon contribution to the total resistivity (the parameter  $\Theta_R$  is considered as a measure of the Debye temperature  $\Theta_D$ ), and the third one accounts for Mott-type interband scattering processes. The least-squares fitting of the above equation to the experimental data of yielded the parameters:  $\rho_0 + \rho_0^{\infty} = 101.7 \,\mu\Omega \,\mathrm{cm}, R = 0.595 \,\mu\Omega \,\mathrm{cm}/\mathrm{K}, \,\Theta_R = 254.1 \,\mathrm{K}$  and  $K = -7.5 \times 10^{-7} \,\mu\Omega \,\mathrm{cm}/\mathrm{K}^3$ .

As the BGM approach neglects crystalline electric field interactions, essentially it is applicable to magnetic systems solely in the temperature range, in which the CEF effect is negligible (i.e. the spin-disorder resistivity can be assumed independent of temperature). In the case of Tb<sub>2</sub>CoGe<sub>6</sub>, the magnetic susceptibility data suggested that the entire <sup>7</sup>F<sub>6</sub> multiplet is thermally well populated above about 100 K. Consequently, the BGM expression was fitted to the experimental data in this high temperature region only (see the solid line in Fig. 2b). The so-derived values of the parameters are as follows:  $\rho_0 + \rho_0^{\infty} = 163.0 \,\mu\Omega \,\text{cm}$ ,  $R = 1.118 \,\mu\Omega \,\text{cm}/\text{K}$ ,  $\Theta_R = 234.1 \,\text{K}$  and  $K = -22.1 \times 10^{-7} \,\mu\Omega \,\text{cm}/\text{K}^3$ . Worth noting is that the value of  $\Theta_R$  is fairly similar to that estimated for Gd<sub>2</sub>CoGe<sub>6</sub>, as expected for isostructural systems, thus proving correctness of the analysis performed.



**Fig. 2.** Temperature dependence of the electrical resistivity of (a)  $Gd_2CoGe_6$  and (b)  $Tb_2CoGe_6$ . The solid lines are the Bloch–Grüneisen–Mott fits described in the text. The insets display for each compound the temperature derivative of the electrical resistivity in the vicinity of the antiferromagnetic phase transition.

The onsets of the antiferromagnetic ordering in Gd<sub>2</sub>CoGe<sub>6</sub> and Tb<sub>2</sub>CoGe<sub>6</sub> manifest themselves as distinct kinks on the  $\rho(T)$  variations at  $T_N = 20$  and 22 K, respectively, which give rise to sharp lambda-like maxima in the temperature derivative of the resistivity (see the insets to Fig. 2). For the Tb-based compound, an additional anomaly in  $d\rho/dT(T)$  is visible near 10 K, and this finding supports

the hypothesis on the order–order transition anticipated from the magnetic susceptibility data (see above). In turn, the resistivity of Gd<sub>2</sub>CoGe<sub>6</sub> rapidly decreases down to the lowest temperatures available in the performed experiment (note large magnitude of  $d\rho/dT$ ), hence subsequent magnetic transition in the ordered state beyond the studied temperature range seems also likely.

#### 4. Conclusions

The two compounds studied in the present work, i.e.  $Gd_2CoGe_6$ and  $Tb_2CoGe_6$ , were found to order antiferromagnetically at  $T_N = 20$  and 22 K, respectively. Moreover, both phases likely exhibit subsequent magnetic transitions at  $T_t < 1.72$  K for the Gd-based germanide and  $T_t \approx 10$  K for the Tb-based one. Regarding this behavior they are quite similar to their Cu-containing counterparts  $Gd_2CuGe_6$  ( $T_N = 20.6$  K,  $T_t < 1.72$  K) and  $Tb_2CuGe_6$ ( $T_N = 33.1$  K,  $T_t = 8.5$  K) [7]. However, in relation to the latter ternaries, the magnetic structures in  $Gd_2CoGe_6$  and  $Tb_2CoGe_6$  may be even more complex because of the anticipated contribution due to the magnetic Co-sublattice. Neutron diffraction experiments are required to determine the magnetic structures in both compounds.

#### References

- [1] O. Sologub, K. Hiebl, P. Rogl, O.I. Bodak, J. Alloys Compd. 227 (1995) 37.
- [2] M. Konyk, L. Romaka, D. Gignoux, D. Fruchart, O. Bodak, Yu. Gorelenko, J. Alloys Compd. 398 (2005) 8.
- [3] O.Ya. Oleksyn, M.F. Fedyna, G.M. Koterlyn, Visnyk L'viv Derzh. Univ. Ser. Khim. 31 (1991) 19.
- [4] M.B. Konyk, P.S. Salamakha, O.I. Bodak, V.K. Pecharsky, Kristallografiya 33 (1988) 838.
- [5] M.B. Konyk, O.I. Bodak, J. Alloys Compd. 387 (2005) 243.
- [6] M. Konyk, L. Akselrud, Yu. Gorelenko, L. Romaka, Visnyk L'viv Univ. Ser. Khim. 49 (2008) 115.
- [7] D. Kaczorowski, M. Konyk, A. Szytuła, L. Romaka, O. Bodak, Solid State Sci. 10 (2008) 1891.
- [8] H. Yamamoto, I. Oguro, M. Ishikawa, J. Phys. Soc. Jpn. 65 (1996) 3464.
- [9] D.X. Li, S. Nimori, Y.X. Li, K. Koyama, Y. Shiokawa, J. Alloys Compd. 408–412 (2006) 122.
- [10] Y.T. Fan, W.H. Lee, Y.Y. Chen, Phys. Rev. B 69 (2004) 132401.
- [11] R. Troć, R. Wawryk, K. Gofryk, A.V. Gribanov, Yu.D. Seropegin, J. Phys.: Condens. Matter 23 (2011) 146001.
- [12] A. Szytula, B. Penc, Yu. Gorelenko, M. Konyk, K. Tomala, A. Winiarski, Chem. Met. Alloys 1 (2008) 88.
- [13] M. Konyk, B. Kuzhel, Yu. Stadnyk, Yu. Gorelenko, Ya. Mudryk, A. Waskiv, J. Alloys Compd. 459 (2008) 18.
- [14] L.G. Akselrud, P.Yu. Zavalii, Yu.N. Grin, V.K. Pecharsky, B. Baumgartner, E. Wolfel, Mater. Sci. Forum 133-136 (1993) 335.
- [15] R.W. Hill, J. Phys. C: Solid State Phys. 19 (1986) 673.
- [16] N.F. Mott, H. Jones, The Theory of the Properties of Metals and Alloys, Oxford University Press, 1958.